Stokes and anti-Stokes Raman spectra of the high energy C-C stretching modes in graphene and diamond

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The Stokes and the anti-Stokes Raman spectra from the high energy carbon-carbon stretching modes in graphene and diamond are studied, with focus on their intensity dependence on the excitation laser power. For diamond, the experiment was performed on a 50μm slice of chemical vapour deposition grown single crystal, using a 785nm laser, both in the continuous wave (CW) and pulsed mode locked (ML – 130fs pulses) excitation regimes. For graphene, the experiments were performed on suspended samples, including single layer, bilayer with AB-Staking, twisted bi-layer, and a sample with 4-6 layers, using a λ=532nm and a λ=633nm laser. The Stokes and anti-Stokes intensities vary largely from sample to sample, and the anti-Stokes/Stokes intensity ratio shows different excitation laser power behaviours. These differences can be explained on the basis of laser heating or correlated Stokes-anti-Stokes scattering phenomena.

1 Introduction Phonons are quantum states of the harmonic oscillators, having their increase and decrease in number n described by creation ($a^\dagger$) and annihilation (a) operators. The action of these operators on an eigenstate $|n\rangle$ introduces normalization factors given by $a^\dagger|n\rangle = (n+1)^{1/2}|n+1\rangle$ and $a^\dagger|n\rangle = n^{1/2}|n-1\rangle$. In the inelastic scattering of light by phonons, a photon creates or annihilates a phonon, in the so-called Stokes or anti-Stokes Raman scattering process. The intensity ratio between the Stoke and anti-Stokes components (I_{aS}/I_S) are, therefore, usually determined by the quantum mechanics normalization factors, $I_{aS}/I_S = n/(n+1)$ [1,2].

The thermal population of phonons is given by the Bose-Einstein distribution function $n_0 = [\exp(E_q/k_BT)-1]^{-1}$, where E_q and k_BT are the phonon and thermal energies, respectively. Replacing n by n_0 in the anti-Stokes/Stokes intensity ratio gives $I_{aS}/I_S = (n_0+1)/n_0 = C\exp(E_q/k_BT)$. This equation can be inverted to obtain the effective phonon temperature in a material by measuring their Stokes and anti-Stokes Raman spectra, given by $T = E_q/\{k_B[\ln C - \ln(I_{aS}/I_S)]\}$. This procedure has been largely used in the literature to study the phonon population dependence of different nanomaterials properties, including electrical power [3], thermal conductivity [4], effective temperatures on biased nanostructures [5], phonon anarmonicities and lifetimes [6,7], and optical transitions [8,9]. Generally, the constant C is found by setting T at room temperature for very low laser power (or zero bias), where sample heating should not be present.

One aspect that is not considered in this type of treatment is the possibility that one single phonon that is created in the Stokes process is subsequently annihilated in the anti-Stokes process. This correlated Stokes-anti-Stokes scattering phenomenon was proposed in 1977 by Klyshko [10], and it has been shown to affect the anti-Stokes and Stokes intensities in graphene [11] and diamond [12,13]. A theoretical framework to address this phenomenon has been proposed, demonstrating implications for the anti-Stokes/Stokes intensity ratio [14].

In this paper we will explore the laser power dependence for diamond and graphene, including the dependence on the layer stacking, on different excitation laser energies, and on the difference between a continuous wave (CW) or a mode locked pulsed (ML) laser regime. Some of the re-

sults shown here have already been discussed in Refs.[11,13,14], and here we present a deeper comparative analysis. The differences can be explained on the basis of laser heating or correlated Stokes-anti-Stokes scattering (SaS) phenomena.

2 Methods

2.1 Samples Graphene and diamond samples are studied here. Graphene has a first-order Raman allowed peak at 1584cm^{-1}, named G band. Diamond has a first-order Raman allowed peak at 1332cm^{-1}. In both cases, these Raman bands are related on the C-C stretching modes.

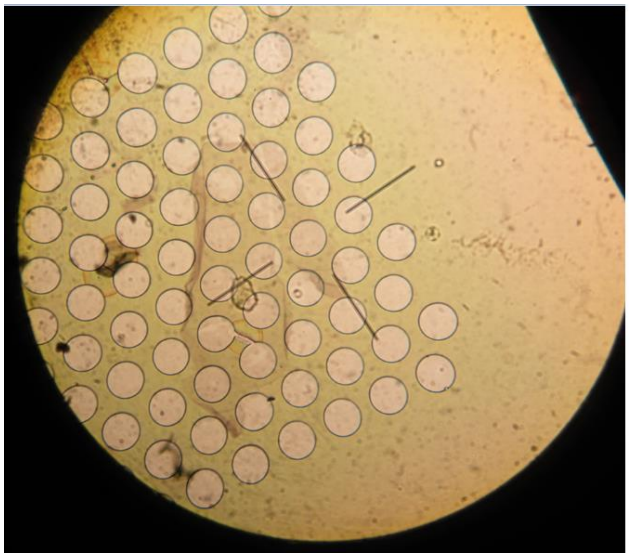

Figure 1 Optical image of the graphene sample. The holes in the SiN_x substrate are 10μm in diameter.

The diamond sample was grown by the chemical vapour deposition (CVD) method, with cutting and polishing generating a highly pure free-standing 50μm thick diamond crystal [13]. As for the graphene samples, Fig.1 shows an optical image. The graphene sample has been deposited on top of a SiN_x substrate with holes. The graphene flakes are suspended over the holes, and the light can pass through the graphene sample without touching the substrate. More details can be found in Ref.[11]. Four types of graphene samples are studied here: single-layer graphene, double-layer graphene with AB-stacking, twisted bilayer graphene (tBLG), and a 4-6 layers graphene with AB-stacking. All the samples were prepared by mechanical exfoliation of graphite, and deposited on a similar substrate, as shown in Fig.1(a). The tBLG [15] was engineered to exhibit a twist angle $\theta = 11°$, which generates a peak in the joint density of states that lies in the energy range of a 633nm and a 532nm lasers [11]. In this way, when exciting the tBLG with a 633nm laser, the G band anti-Stokes Raman emission is resonant. When exciting the tBLG sample with the 532nm laser, the G band Stokes Raman emission is resonant.

2.2 Experimental details The graphene samples were excited with a frequency doubled Nd:YVO_4 laser at E_{laser} = 2.33 eV (λ_{laser} = 532.0 nm) and with a HeNe laser at E_{laser} = 1.96 eV (λ_{laser} = 632.8 nm), both working in the continuous wave (CW) regime. As already specified, these two lasers are off resonance with respect to the peak in the joint density of states (JDOS) for the tBLG, E_{laser} being blue- or red-shifted from the JDOS peak by approximately the G band phonon energy (E_G = 0.2 eV) [11]. The signal was collected through a spectrometer, equipped with a charge coupled device (CCD) detector, and filtered using optical notches and band pass filters.

The diamond sample was studied with a Ti:Sapph laser at λ = 785nm, which can work both in the CW (continuous wave) and ML (mode locked – pulsed) regimes. For ML, the duration of the excitation laser pulses is τ = 130fs, with a repetition rate of Δf = 76MHz.

Finally, a 130fs pulsed laser at λ=545nm from an OPO was used to excite both diamond and one of the graphene samples, for comparative analysis.

3 Results and discussions

3.1 Pulsed laser Raman spectra Figure 2 shows the Stokes and anti-Stokes Raman spectra of the 4-6 layers graphene and diamond samples, obtained with a 130fs pulsed laser at λ=545nm, with average excitation laser power P_{laser} = 20mW. For diamond (olive lines), the central peak at λ=545nm is leakage from the laser, and the peaks at 512nm and 592nm are the anti-Stokes and Stokes Raman peaks, respectively. The width of the peaks is determined by the spectral width of the 130fs laser pulses.

For graphene (light green in Fig.2), again the peak at λ=545nm is leakage from the laser, while the anti-Stokes and Stokes Raman peaks appear at wavelengths (505nm and 602nm, respectively), which are different from those in diamond, since the phonon energy is different. To measure the Raman signals, we used a band-pass filter at (506±20)nm for the anti-Stokes signal, and a band-pass filter at (600±20)nm for the Stokes signal. The square-shaped signals observed within these ranges in Fig.2 come from the hot luminescence of graphene, which appears when graphene is excited with pulsed lasers [16].

Analysing the Raman peak intensities with a single Lorentizan fit, we found the Stokes/anti-Stokes intensity ratio for diamond I_S/I_{aS} = 630, which is close to the expected value for the diamond phonon (1332cm^{-1}), in agreement with the Bose-Einstein phonon population at room temperature (T = 295K). For the graphene sample (G band at 1584cm^{-1}), the Stokes/anti-Stokes intensity ratio is I_S/I_{aS} = 5.3, which is three orders of magnitude smaller

than the expected value from the Bose-Einstein phonon population at room temperature. This result indicates that either the graphene sample exhibits a very high laser-induced phonon temperature, or that the Stokes-anti-Stokes correlated scattering phenomenon is the dominating process. Correlation measurements, as performed in Refs.[12,13] are not possible here due to the strong hot luminescence [16].

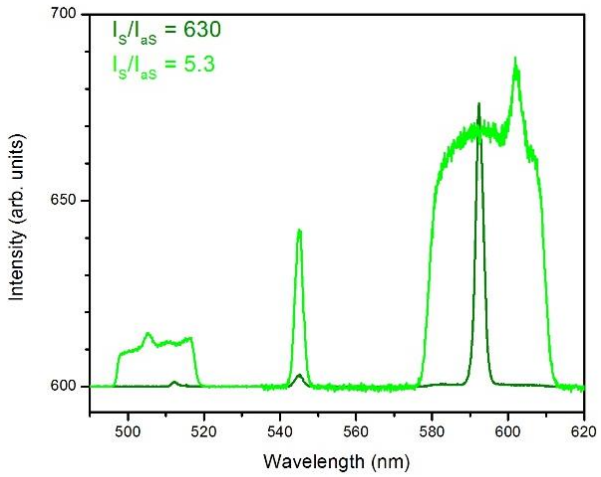

Figure 2 Stokes and anti-Stokes Raman spectra of graphene (light green) and diamond (olive) obtained with a 130fs pulsed laser at λ=545nm. The Stokes/anti-Stokes intensity ratio (I_S/I_{aS}) is also displayed for both samples, following the same colour code. The 1332cm^{-1} diamond peak appear at 592nm and 512nm for the Stokes and anti-Stokes scattering, respectively. The 1584cm^{-1} graphene peak appears at 602nm and 505nm for the Stokes and anti-Stokes scattering, respectively.

3.2 Spectral power dependence Figure 3 shows the excitation laser power dependence for the Stokes and anti-Stokes intensities normalized by the excitation laser power, from all the samples studied in this work. Green, red and black/grey symbols indicate Raman spectra obtained with the lasers at λ_{laser} = 532.0 nm, λ_{laser} = 632.8nm and λ_{laser} = 785 nm, respectively. The different symbol shapes indicate the specific samples, as displayed in the figure legend. Graphene data were obtained with CW lasers. Filled and opened symbols stand for Stokes and anti-Stokes intensities, respectively, except for diamond with the CW laser, where black and grey stand for Stokes and anti-Stokes intensities, respectively.

In Fig.3, the laser-power normalized Stokes intensities (filled symbols) show a constant value, in agreement with a linear dependence of I_S on the excitation laser power. The laser-power normalized anti-Stokes intensities exhibit a super-linear power dependence. Therefore, the anti-Stokes responses demonstrate that either laser-induced heating or correlated Stokes-anti-Stokes scattering are taking place.

Figure 4 shows more clearly the behaviour of the diamond Raman responses. For CW excitation, both Stokes (black spheres) and anti-Stokes (grey spheres) exhibit a linear dependence with the excitation laser power, which is evident in Fig.4 by a constant value for "Intensity/Laser power". For the ML pulsed excitation, the Stokes spectra (black circles) shows a linear dependence on the excitation laser power, evidenced by a constant "Intensity/Laser power" in Fig.4, while the anti-Stokes (open circles) shows a super-linear dependence.

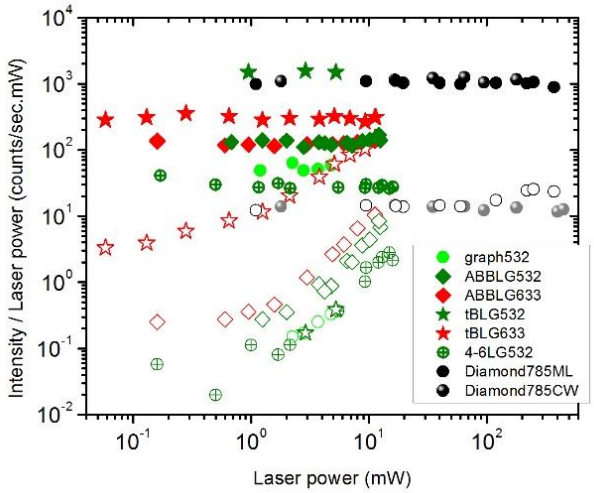

Figure 3 Laser power dependence for the Stokes and anti-Stokes scattering intensities for diamond and different graphene samples. The different samples are indicated by different symbols, according to the legend: graph532 – single layer graphene excited with λ=532nm; ABBLG532 – AB-stacked bilayer graphene excited with λ=532nm; ABBLG633 – AB-stacked bilayer graphene excited with λ=633nm; tBLG532 – twisted bilayer graphene excited with λ=532nm; tBLG633 – twisted bilayer graphene excited with λ=633nm; 4-6LG532 – 4 to 6-layers graphene with AB-stacking excited with λ=532nm (graphene data were obtained with CW lasers); Diamond785ML – diamond excited with λ=785nm in the pulsed mode locked regime; Diamond785CW – diamond excited with λ=785nm in the continuous wave regime.

3.3 anti-Stokes/Stokes intensity ratio Figure 5 shows the excitation power dependence behaviour for the I_{aS}/I_S intensity ratio for all the samples studied in this work.

For diamond, I_{aS}/I_S is constant for the CW excitation (black circles). This indicates that neither heating nor the correlated Stokes-anti-Stokes scattering phenomena are relevant [17]. However, for the pulsed ML excitation, the behaviour is super-linear. It has been shown already that this super-linear behaviour is due to the correlated Stokes-anti-Stokes scattering (SaS) process [12-14]. Since the SaS is a non-linear process, the ultra-high intensities within a

pulse in the ML pulsed laser regime are needed to produce the effect in diamond.

For graphene, a difference in behaviour is observed when comparing the tBLG in resonance with the anti-Stokes photon emission (red star data) and all the other samples. The data are fit with the theory introduced in Ref.[14]. The fittings of the excitation laser power dependences for single layer graphene (light green circle), AB-stacked bilayer graphene (diamond symbols), and 4-6-layers graphene (crossed circles) indicate an I_{aS}/I_S dependence dominated by thermal effects. The fitting of the excitation laser power dependence for twisted bilayer graphene in resonance with the anti-Stokes photon emission (red star data) indicates an I_{aS}/I_S dependence dominated by the SaS correlated scattering, consistent with Ref.[14]. For tBLG in resonance with the Stokes photon emission (green star data), only two data points are obtained due to the very low anti-Stokes intensity, which is not enough data for a fitting analysis. For graphene all the spectra where obtained in the CW excitation regime.

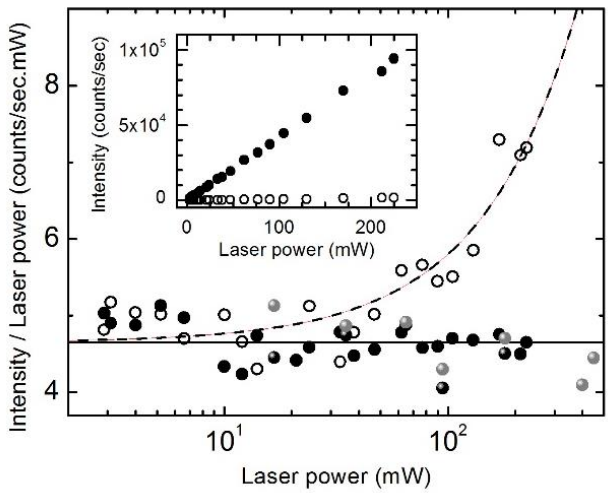

Figure 4 Excitation laser power dependence for the Stokes and anti-Stokes Raman intensities for diamond, normalized (1) by the laser power and (2) by a constant factor to merge the Stokes and anti-Stokes intensities. Black and grey spheres stand for Stokes and anti-Stokes Raman intensities obtained in the CW excitation regime. Filled and opened circles stand for the Stokes and anti-Stokes Raman intensities obtained in the pulsed ML excitation regime. For the spectra obtained in the ML pulsed regime, laser power means the average laser power, and not the peak laser power. The inset shows the Stokes and anti-Stokes intensities as measured in the pulsed ML regime, in a linear scale.

When comparing graphene and diamond, the first important difference is the phonon energy, which influences on the Bose-Einstein distribution. Additionally, diamond is transparent for visible and lower light energies, while graphene is always in resonance with the light absorption and emission. The SaS phenomenon appears here in diamond when ultra-high intensities from a pulsed laser is in place, while the specially engineered tBLG shows the phenomenon at much lower average powers, even for a CW laser. More work is needed to characterize the SaS process in graphene, especially because the strong hot luminescence prevents the use of pulsed lasers for photon coincidence experiments.

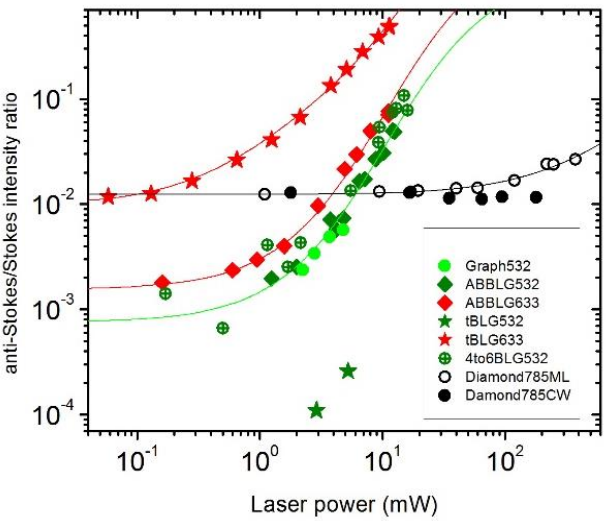

Figure 5 Excitation laser power dependence for the anti-Stokes/Stokes Raman intensity ratio (I_{aS}/I_S) for diamond (black data) and for the different graphene samples (green and red data). Sample specification is found in the figure legend: graph532 – single layer graphene excited with λ=532nm; ABBLG532 – AB-stacked bilayer graphene excited with λ=532nm; ABBLG633 – AB-stacked bilayer graphene excited with λ=633nm; tBLG532 – twisted bilayer graphene excited with λ=532nm; tBLG633 – twisted bilayer graphene excited with λ=633nm; 4-6BLG532 – 4 to 6-layers graphene with AB-stacking excited with λ=532nm; Diamond785ML – diamond excited with λ=785nm in the pulsed mode locked regime; Diamond785CW – diamond excited with λ=785nm in the continuous wave regime. The lines are fitting to the data according with the theory presented in Ref.[14].

3 Conclusions In this work we measured the power dependence of the Stokes and anti-Stokes intensities from a 50μm thick diamond crystal and from different graphene samples, including single layer, bilayer and 4 to 6-layers graphenes. The I_{aS}/I_S intensity ratio varies from sample to sample, and it also depends on the excitation regime (pulsed vs. continuous wave). The results are analysed considering two distinct effects, namely laser induced heating and the production of correlated Stokes-anti-Stokes photon pairs.

Our results indicate that there is no heating in Diamond. For CW laser excitation, this is proven by a constant value of the I_{aS}/I_S ratio. For the pulsed laser excitation, the increase in the I_{aS}/I_S ratio comes from the correlated Stokes-anti-Stokes phenomenon. For graphene, the correlated Stokes-anti-Stokes phenomena is dominant for tBLG when

in resonance with the scattered anti-Stokes emission. Otherwise, heatting is the dominant phenomenon.

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